

# CIVIL-408

## Multiscale Modeling in Mechanics

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### Week 11

Which of the following are key characteristics of **Data-Driven Computational Mechanics (DDCM)**? Select all that apply.

- a. It replaces constitutive models with a material dataset.
- b. It enforces equilibrium and compatibility through numerical solvers.
- c. It requires training a neural network on large datasets.
- d. It projects strain–stress states onto the closest admissible data point in phase space.
- e. It extrapolates well beyond the range of provided data.

Which statements accurately describe **ML-based surrogate modeling** in solid mechanics? Select all that apply.

- a. It can approximate nonlinear constitutive behavior using neural networks
- b. It often requires explicit enforcement of physical constraints such as symmetry or objectivity.
- c. It may struggle with extrapolation outside the training domain.
- d. It is inherently unable to model path-dependent behavior.

## Homogenization for structural elements

- Beams
- Shells

## Homogenization for structural materials

- Concrete
- Wood
- Cellular/architected solids

## Homogenization for geomaterials

- Jointed rocks
- Granular materials
- Clays

## Homogenization for structural elements

- Beams
- Shells

## Homogenization for structural materials

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- Cellular/architected solids

## Homogenization for geomaterials

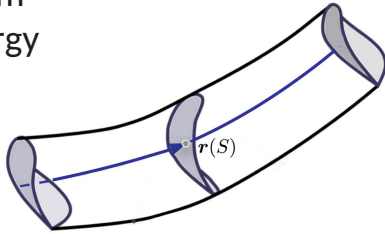
- Jointed rocks
- Granular materials
- Clays

# Homogenization for structural elements

## Beams under finite strains

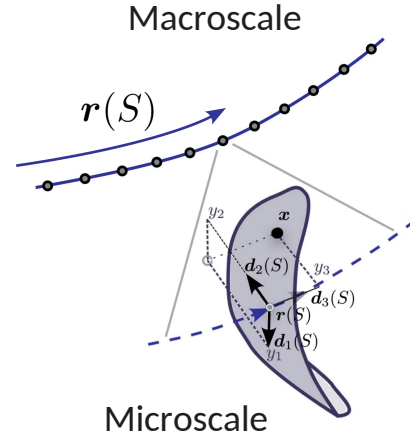
Continuum strain energy

$$\psi(\mathbf{F})$$



Asymptotic  
dimension reduction

Le Clezio et al, 2023

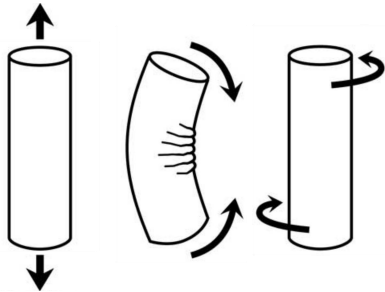


Condensed beam energy

$$\psi_{\text{eff}}(\varepsilon, \kappa_1, \kappa_2, \tau)$$

$\mathbf{E}$

Beam kinematics

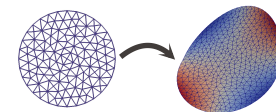


Stretching  $\varepsilon(S)$     Bending  $\kappa_1(S) \kappa_2(S)$     Twist  $\tau(S)$

$$\mathbf{F} = \mathbf{F}(\mathbf{E}(S), \mathbf{y}(X, S)) \rightarrow \min_{\mathbf{y}} \int_0^L \int_{\mathcal{A}} \psi(\mathbf{F}(\mathbf{E}, \mathbf{y})) d\mathcal{A} dS \rightarrow \psi_{\text{eff}}(\mathbf{E})$$

Microscopic coordinates

Condense out (FE<sup>2</sup>)



## Shells

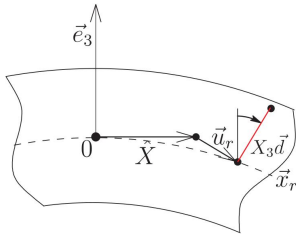
**Macroscale** (Mindlin-Reisner shell)

Kinematics:

In-plane strains/Transverse shear

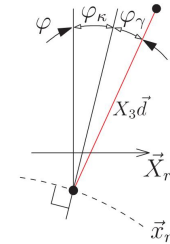
$$\varepsilon_{\alpha\beta}$$

$$\gamma_{\alpha}$$



Out-of-plane curvature

$$\kappa_{\alpha\beta}$$



Assumption: Directors remain straight and are inextensible

Special care needs to be taken for thick shells!

**Microscale** (3D continuum)

Enforcing macroscale averages:

$$u^z(x_\beta^r, z) = -\frac{1}{2}x_\alpha^r \kappa_{\alpha\beta} x_\beta^r + \tilde{u}^z$$

$$u_\alpha^r(x_\beta^r, z) = \varepsilon_{\alpha\beta} x_\beta^r + z \kappa_{\alpha\beta} x_\beta^r + \tilde{u}_\alpha^r$$

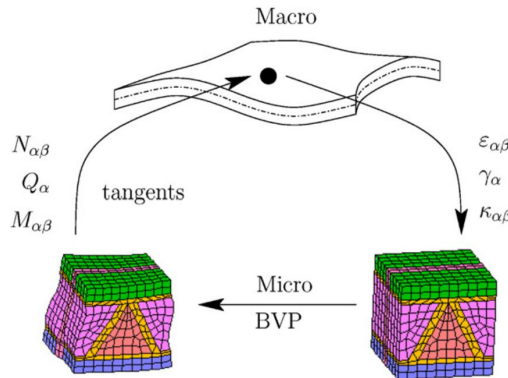
Periodic fluctuations subject to kinematic constraints

Computing macroscale generalized forces:

$$N_{\alpha\beta} = \int_{-H/2}^{H/2} \sigma_m^{\alpha\beta} dh$$

$$V_\alpha = \frac{1}{2} \int_{-H/2}^{H/2} (\sigma_m^{3\alpha} + \sigma_m^{\alpha 3}) dh$$

$$M_{\alpha\beta} = \int_{-H/2}^{H/2} h \sigma_m^{\alpha\beta} dh$$



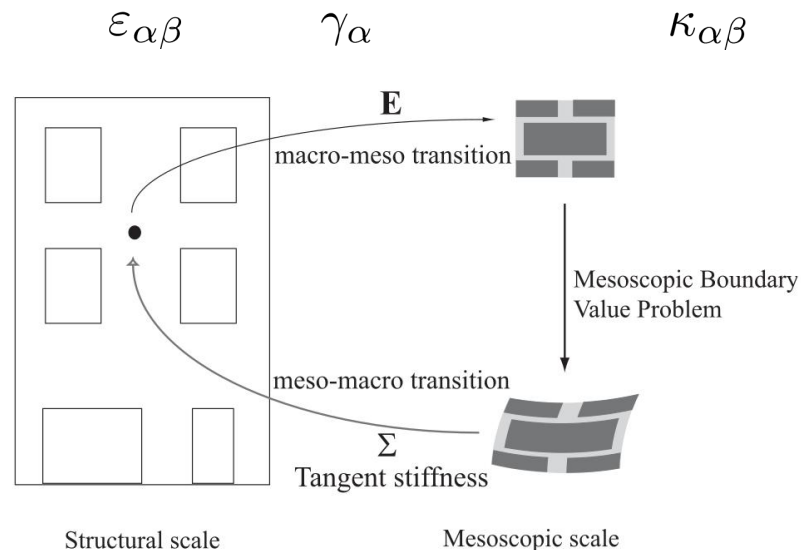
## Masonry flat shells - Case of *homogeneous* deformation

**Macroscale** (Mindlin-Reisner shell)

Kinematics:

In-plane strains/Transverse shear

Out-of-plane curvature



**Microscale** (3D continuum)

Energy consistency between scales:

$$N_{\alpha\beta}\varepsilon_{\alpha\beta} + M_{\alpha\beta}\kappa_{\alpha\beta} + V_{\alpha}\gamma_{\alpha}$$

$$= \frac{1}{V} \int_V \sigma_{ij}\varepsilon_{ij}dV$$

What if there is **bifurcation** from a homogeneous deformation state?

Then it should be detected, and a new upscaling procedure is needed that accounts for the correct dissipated energy/mode.

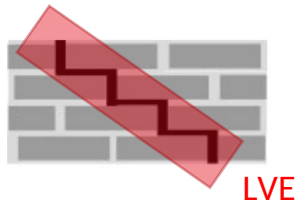
## Masonry flat shells - Case of *localized* deformation

The bifurcation from a macroscopically homogeneous deformation state manifests itself as a discrete localized band. Generally hierarchical techniques have trouble accounting for true localized displacement fields.

Periodic



True/Realistic



Energy consistency between scales needs to account now also the localized contribution.

Localization can be handled by appropriate kinematic enrichment at the macroscale, where the jump happens across a cohesive zone:

$$u^r = u^r + [[u_d^r]] \Psi^r \quad \leftarrow \begin{array}{l} \text{unit jumps} \\ \text{along} \\ \text{discontinuity} \\ \text{lines} \end{array}$$

$$u^z = u^z + [[u_d^z]] \Psi^z \quad \leftarrow \begin{array}{l} \text{unit jumps} \\ \text{along} \\ \text{discontinuity} \\ \text{lines} \end{array}$$

Can obtain new generalized strains accounting for the discontinuities

$$\varepsilon_{\alpha\beta} \quad \gamma_\alpha \quad \kappa_{\alpha\beta}$$

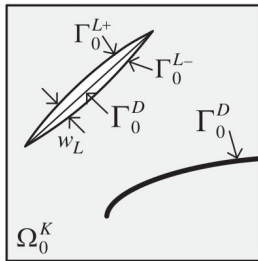
At the microscale, one needs to consider also a localized volume element (LVE).

## Overcoming the lack scale separation

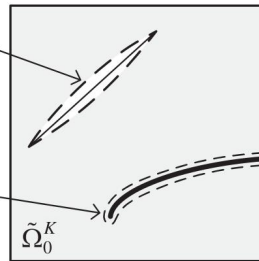
The previous example represents a case of a more general continuous-discontinuous homogenization approach in the presence of interfaces/cracks, e.g. via the Multiscale Aggregating Discontinuities method.

### Microscale

A material instability is detected (loss of ellipticity)

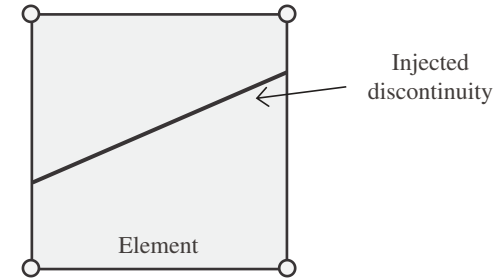


exclude localization band  
exclude set of measure zero



### Macroscale

Inject discontinuity to the macroscale guaranteeing material stability (ellipticity)

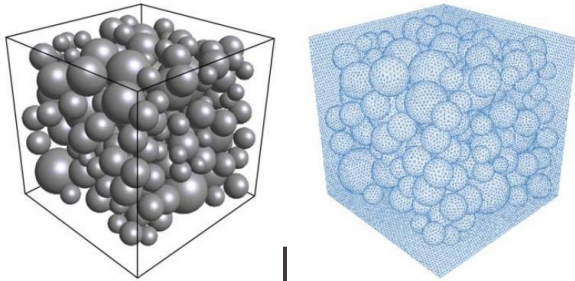


- Compute stress on a modified “perforated” unit cell that excludes cracks  $\langle \mathbf{P}^K \rangle = \frac{1}{|\tilde{\Omega}_0^K|} \int_{\tilde{\Omega}_0^K} \mathbf{P}^K \, d\Omega$

- Compute orientation/magnitude of discontinuity  $(\bar{\mathbf{U}}^{K-1}, \bar{\mathbf{N}}^{K-1}) = \arg \left( \min_{\bar{\mathbf{U}}, \bar{\mathbf{N}}} (\bar{\mathbf{U}} \otimes \bar{\mathbf{N}} - \mathcal{F}^{K-1} + \langle \mathbf{F}^K \rangle)^2 \right)$

## A damage-based approach

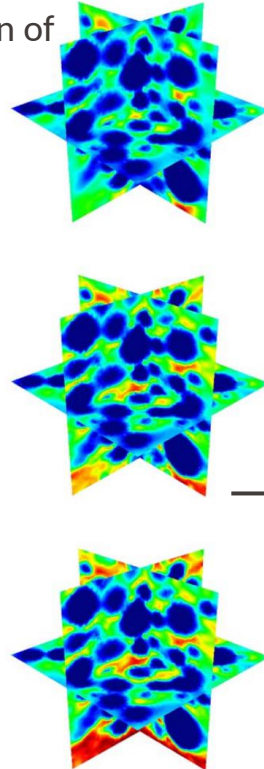
Mesostructure generation  
(generation of aggregates and their placement within the matrix)



FEM simulations using  
damaging material

$$\begin{aligned}\boldsymbol{\sigma} &= (1 - D)\mathbb{C}\boldsymbol{\varepsilon} \\ \rho\psi &= \frac{1}{2}\boldsymbol{\sigma} = (1 - D)\boldsymbol{\varepsilon}\mathbb{C}\boldsymbol{\varepsilon} \\ \dot{\phi} &= -\rho\frac{\partial\psi}{\partial D}\dot{D}\end{aligned}$$

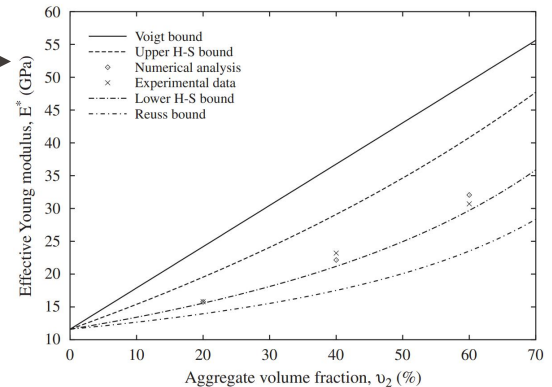
Distribution of  
damage  
parameter



Calculation of effective stiffness

$$\begin{Bmatrix} \langle \sigma_{11} \rangle_{\Omega} \\ \langle \sigma_{22} \rangle_{\Omega} \\ \langle \sigma_{33} \rangle_{\Omega} \\ \langle \sigma_{12} \rangle_{\Omega} \\ \langle \sigma_{23} \rangle_{\Omega} \\ \langle \sigma_{31} \rangle_{\Omega} \end{Bmatrix} = \begin{bmatrix} C_{11}^* & C_{12}^* & C_{13}^* & C_{14}^* & C_{15}^* & C_{16}^* \\ C_{21}^* & C_{22}^* & C_{23}^* & C_{24}^* & C_{25}^* & C_{26}^* \\ C_{31}^* & C_{32}^* & C_{33}^* & C_{34}^* & C_{35}^* & C_{36}^* \\ C_{41}^* & C_{42}^* & C_{43}^* & C_{44}^* & C_{45}^* & C_{46}^* \\ C_{51}^* & C_{52}^* & C_{53}^* & C_{54}^* & C_{55}^* & C_{56}^* \\ C_{61}^* & C_{62}^* & C_{63}^* & C_{64}^* & C_{65}^* & C_{66}^* \end{bmatrix} \begin{Bmatrix} \langle \varepsilon_{11} \rangle_{\Omega} \\ \langle \varepsilon_{22} \rangle_{\Omega} \\ \langle \varepsilon_{33} \rangle_{\Omega} \\ 2\langle \varepsilon_{12} \rangle_{\Omega} \\ 2\langle \varepsilon_{23} \rangle_{\Omega} \\ 2\langle \varepsilon_{31} \rangle_{\Omega} \end{Bmatrix}$$

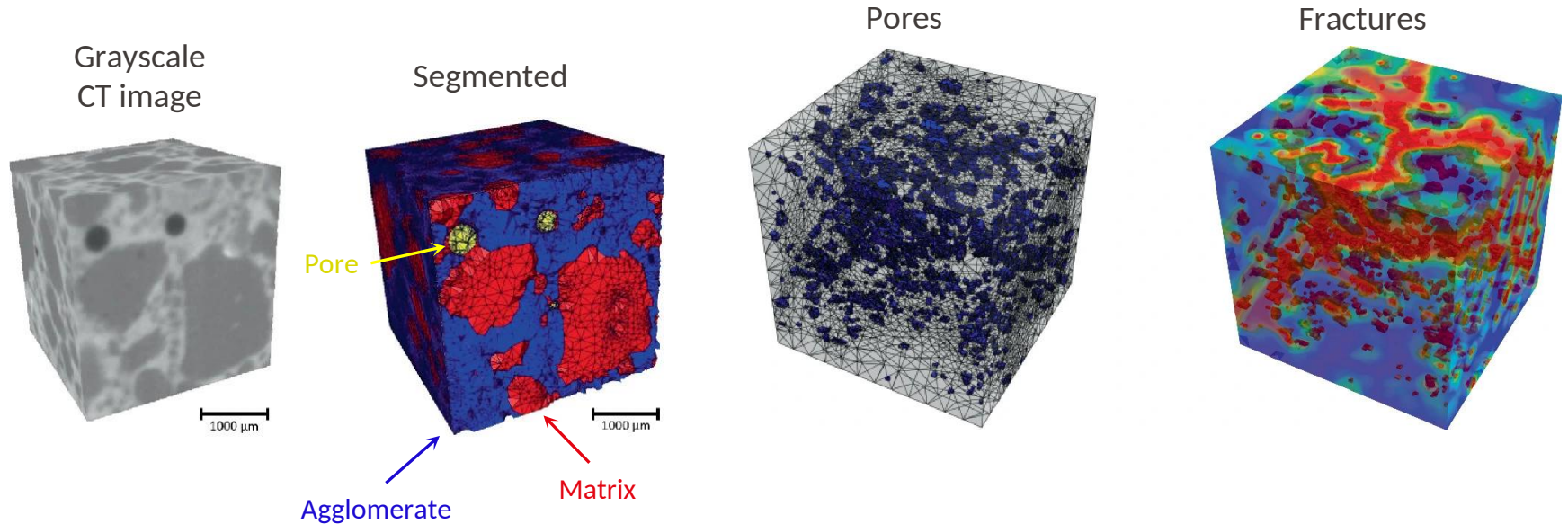
Comparison against experiments



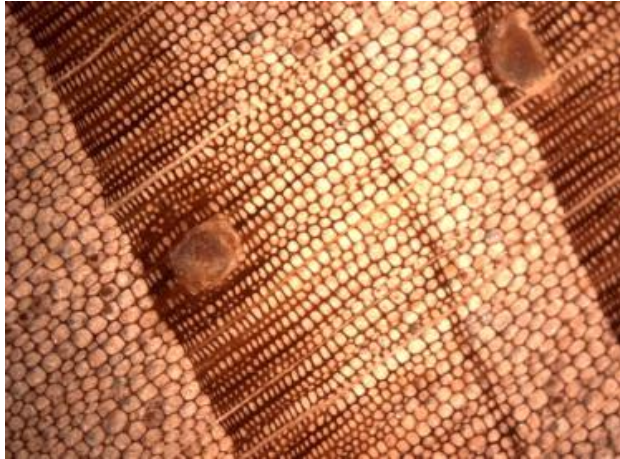
# Homogenization for concrete

## A damage-based approach

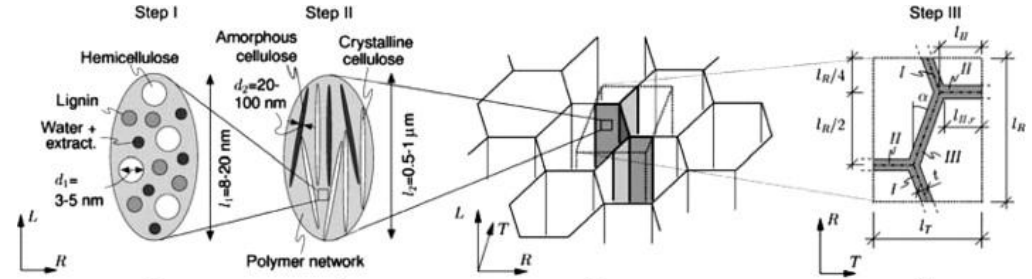
Studies also apply these ideas to digital twins from X-ray Computed Tomography



Cellular microstructure revealed from micrograph of structural wood



Three-step homogenization procedure



$$\mathbb{C}_{\text{polynet}}^{\text{SCI}} = \left\{ \sum_r \tilde{f}_r \mathbb{C}_r : [\mathbb{I} + \mathbb{P}_{\text{sph}}^{\text{polynet}} : (\mathbb{C}_r - \mathbb{C}_{\text{polynet}}^{\text{SCI}})]^{-1} \right\} : \left\{ \sum_s \tilde{f}_s [\mathbb{I} + \mathbb{P}_{\text{sph}}^{\text{polynet}} : (\mathbb{C}_s - \mathbb{C}_{\text{polynet}}^{\text{SCI}})]^{-1} \right\}^{-1}$$

$$\mathbb{C}_{\text{cwm}}^{\text{MTI}} = \left\{ f_{\text{polynet}} \mathbb{C}_{\text{polynet}}^{\text{SCI}} + \sum_r f_r \mathbb{C}_r : [\mathbb{I} + \mathbb{P}_{\text{cyl}}^{\text{polynet}} : (\mathbb{C}_r - \mathbb{C}_{\text{polynet}}^{\text{SCI}})]^{-1} \right\}$$

$$: \left\{ f_{\text{polynet}} \mathbb{I} + \sum_s f_s : [\mathbb{I} + \mathbb{P}_{\text{cyl}}^{\text{polynet}} : (\mathbb{C}_s - \mathbb{C}_{\text{polynet}}^{\text{SCI}})]^{-1} \right\}^{-1}, \quad r, s \in [\text{crycel}, \text{amocel}].$$

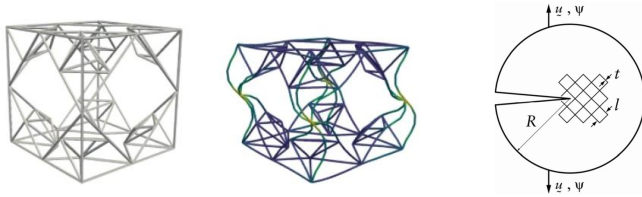
$$\mathbb{C}_{\text{SW}}^{\text{MTII}} = \{(1 - \hat{f}_{\text{lum}}) \mathbb{C}_{\text{cwm}}^{\text{MTI}}\} : \{(1 - \hat{f}_{\text{lum}}) \mathbb{I} + \hat{f}_{\text{lum}} : [\mathbb{I} - \mathbb{P}_{\text{cyl}}^{\text{cwm}} : \mathbb{C}_{\text{cwm}}^{\text{MTI}}]^{-1}\}^{-1}$$

## The future of structural materials

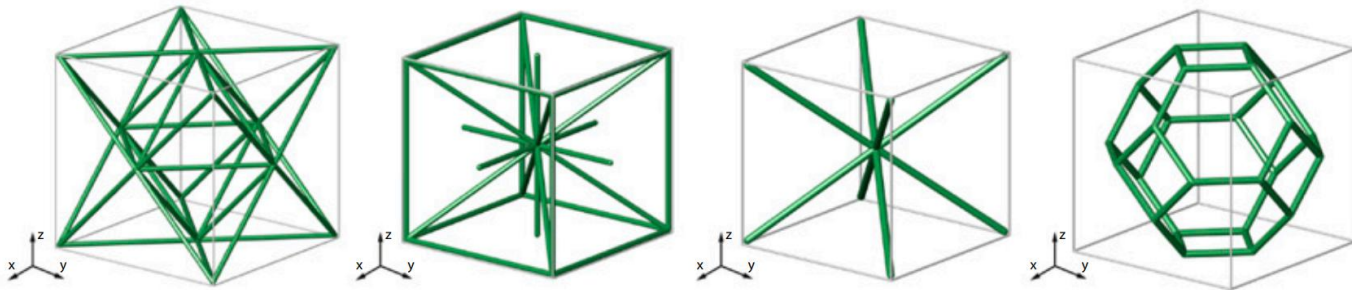
Lattice materials furnish nowadays a manufacturable solution to near-optimal structural material properties (stiff, light, strong and damage-tolerant)

Homogenization of single unit cells composed of beam elements:

Can be used to derive scaling laws of their properties

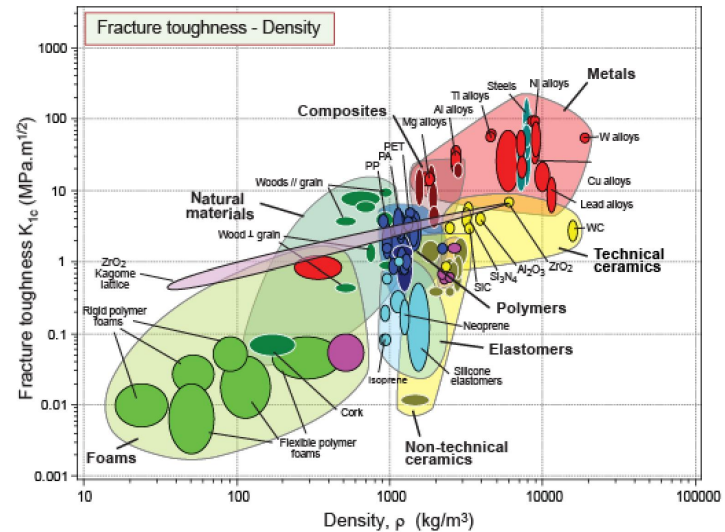
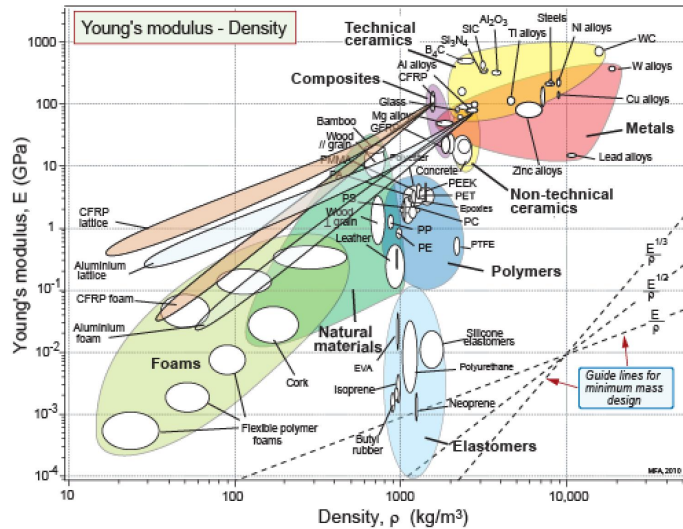


Elasticity:  $E/E_s = B\bar{\rho}^b$   
 Strength:  $\sigma_C/\sigma_{cs} = C\bar{\rho}^c$   
 Fracture toughness:  $K_{IC}/\sigma_{cs}\sqrt{l} = D\bar{\rho}^d$



# Homogenization for architected/cellular solids

Expanding the boundaries of material property space



## Homogenization for structural elements

- Beams
- Shells

## Homogenization for structural materials

- Concrete
- Wood
- Cellular/architected solids

## Homogenization for geomaterials

- Jointed rocks
- Granular materials
- Clays

## Homogenization of jointed-rock mass

Effective medium theory under non-interacting assumptions may be used to capture rock elasticity. The effective compliance tensor is represented as the sum of a background compliance tensor and contributions related to individual non-interacting fractures

Rock:

$$\boldsymbol{\varepsilon} = \mathbf{s}^m : \boldsymbol{\sigma}$$

Joints:

$$[\boldsymbol{\xi}] = \mathbf{s}^j : \mathbf{T}, \quad \mathbf{T} = \boldsymbol{\Sigma} \cdot \mathbf{n}^j$$

Total strain derived from the bulk and the joints

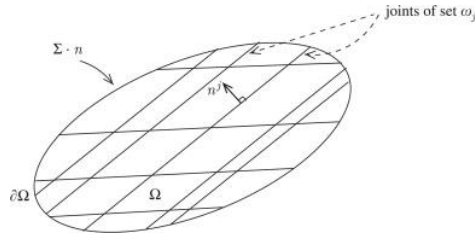
$$\mathbf{E} = \langle \boldsymbol{\varepsilon} \rangle + \frac{1}{|\Omega|} \int_{\Omega} \mathbf{n} \otimes [\boldsymbol{\xi}] dS$$

$$\mathbf{E} = \mathbf{s}^m : \boldsymbol{\Sigma} + a^j (\mathbf{s}^j \cdot \boldsymbol{\Sigma} \cdot \mathbf{n}^j) \otimes \mathbf{n}^j$$

Rewritten as

$$\mathbf{E} = \mathbf{S}^{\text{hom}} : \boldsymbol{\Sigma}$$

$$\mathbf{S}^{\text{hom}} = \mathbf{s}^m + a^j s_{\alpha\beta}^j \mathbf{e}_{\alpha}^j \otimes \mathbf{n}^j \otimes \mathbf{e}_{\beta}^j \otimes \mathbf{n}^j$$



Can be extended to an elastoplastic formulation (plastic sliding of joints)

# Application to geomaterials

## FEMxDEM for localization in granular materials

With appropriate choice of RVEs, FEMxDEM may also simulate shear banding in these materials

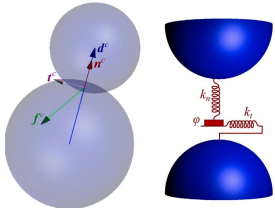
### Microscale (RVE)

Love-Weber formula

$$\sigma = \frac{1}{V} \sum_c l^c \otimes f^c$$

Tangent operator

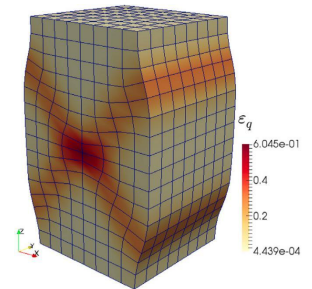
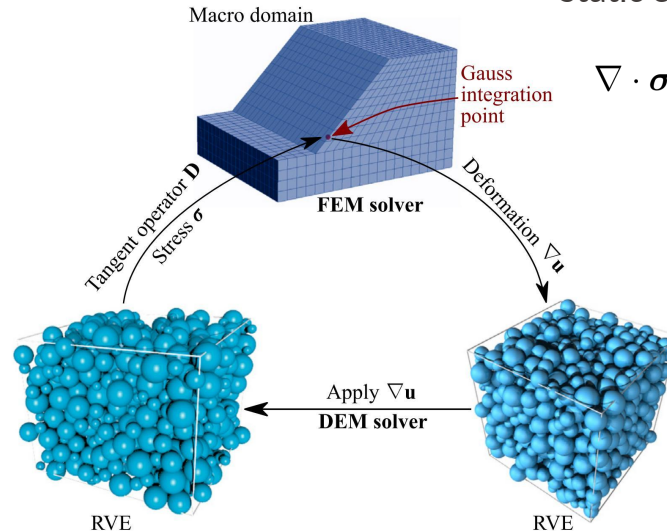
$$C = \frac{1}{V} \sum_c k_n n^c \otimes l^c \otimes n^c \otimes l^c + \frac{1}{V} \sum_c k_t t^c \otimes l^c \otimes t^c \otimes l^c$$



### Macroscale (Continuum)

Static equilibrium

$$\nabla \cdot \sigma + \rho b = 0$$



Guo and Zhao, 2016 Comp. Geotech.

## Granular micromechanics

Analytical homogenization techniques furnish a useful efficient alternative beyond FEMxDEM.

**Contact scale**

Relative displacement  
between two particles (m,n):

$$\delta_i^{mn} = u_i^m - u_i^n + e_{ijk}(\omega_j^m r_k^m - \omega_j^n r_k^n)$$

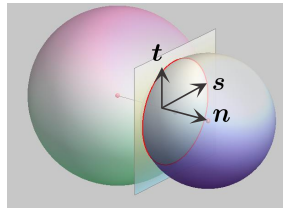
Rewrite using mean strain and  
fluctuation (w/ equilibrium closure):

$$\delta_i^{mn} = (\delta_{jl} l_k^{lmn} - \Gamma_{jkl}^n) \varepsilon_{kl}^n$$

Ensuing contact force:

$$f_i = K_{ij} \delta_j$$

$$K_{ij} = K_n n_i n_j + K_s (s_i s_j + t_i t_j)$$

**Cell scale**

Split system into cells s.t.  
the stress of each cell:

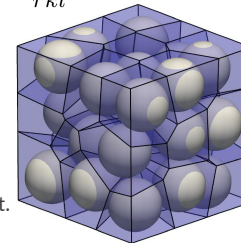
$$\sigma_{ij}^n = \frac{1}{2V^n} \sum_m l_i^{nm} f_j^{nm}$$

relates to the cell strain via:

$$\sigma_{ij}^n = C_{ijkl}^n \varepsilon_{kl}^n$$

Combining everything:

$$C_{ijkl}^n = \frac{1}{2V^n} \sum_m l_i^{nm} K_{jl}^{nm} l_k^{nm} - \frac{1}{2V^n} \sum_m l_i^{nm} K_{jr}^{nm} \Gamma_{rkl}^n$$

**Assembly (macro) scale**

By volume averaging

$$\bar{\sigma}_{ij} = \frac{1}{V} \sum_n V^n \sigma_{ij}^n$$

$$\bar{\varepsilon}_{ij} = \frac{1}{V} \sum_n V^n \varepsilon_{ij}^n$$

Derive concentration  
tensor by solving an  
Eshelby-like problem:

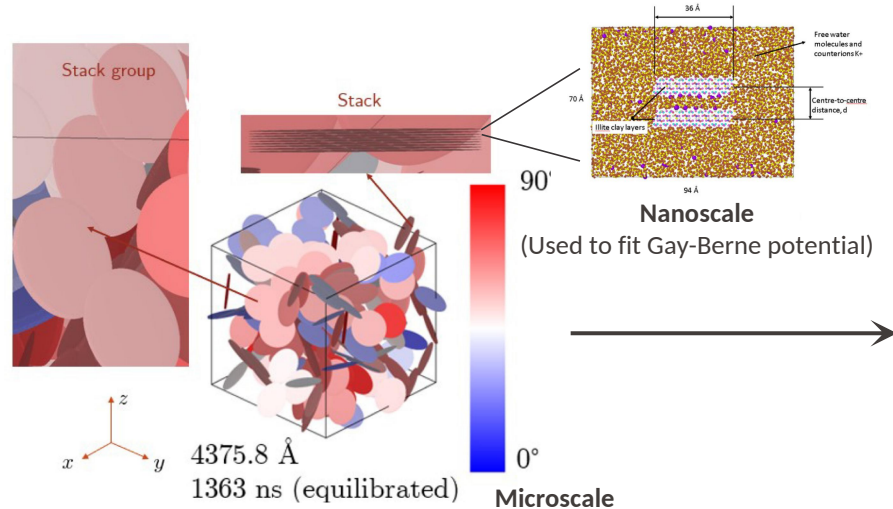
$$\varepsilon_{kl}^n = H_{mnlk}^n \bar{\varepsilon}_{kl}$$

Obtain in the end:

$$\begin{aligned} \bar{\sigma}_{ij} &= C_{ijkl} \bar{\varepsilon}_{kl} \\ &= \left( \frac{1}{V} \sum_n V^n C_{ijmn}^n H_{mnlk}^n \right) \bar{\varepsilon}_{kl} \end{aligned}$$

## Homogenization for clays

Coarse-grained Molecular Dynamics simulations, with fitted MD potential from nanoscale models



## Macroscale

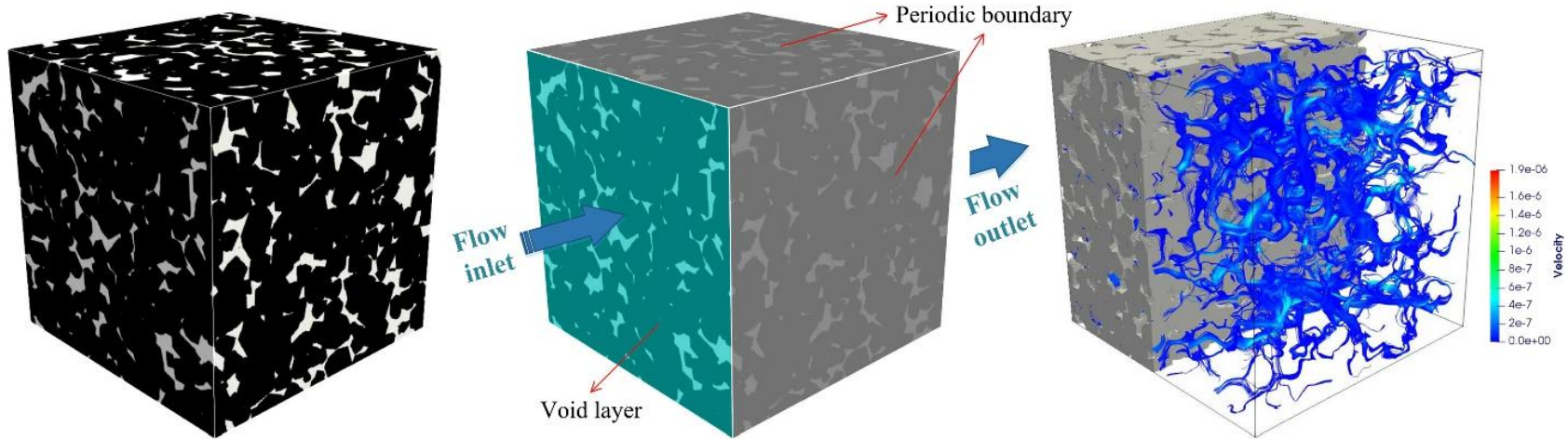
The consider a macroscopic model where the energy of the system  $U$  is divided into:

- *interstack* potential energy
- *intrastack* potential energy

These are derived from system parameters (stack density, average stack size, interstack pair correlation function, fabric tensor, ...)

Rarely are these materials interacting only mechanically. Multiphysics interactions matter, e.g.:

## Poromechanical effects



**That's what I prepared for you today.**

**What would you like to discuss?**

# Reading for next class:

Collection of research articles  
**(Week 12 - Reading Assignments)**